

Antimonides with $D8_8$ and Hf_5Sn_3Cu Structure Types

BY W. RIEGER AND E. PARTHÉ

*School of Metallurgy and Materials Science and Laboratory for Research on the Structure of Matter,
University of Pennsylvania, Philadelphia, Pennsylvania, U.S.A.*

(Received 28 February 1967)

The crystal structures of eight binary rare earth antimonides of composition R_5Sb_3 are found to be of the $D8_8$ structure type. Seven ternary compounds of composition R'_5Sb_3X (where $R' = Ti, Zr, Hf$ and $X = Ni, Cu, Zn$) select the Hf_5Sn_3Cu structure type, a filled up version of the $D8_8$ structure type.

In continuation of an intensive research program (Jeitschko & Parthé, 1967; Jeitschko & Parthé, 1965; Schob & Parthé, 1964), to investigate the occurrence and stability of the $D8_8$ structure type, we studied the crystal structures of binary rare earth antimonides of composition R_5Sb_3 and Ni, Cu-, Zn-stabilized antimonides of Ti, Zr and Hf of composition $(Ti, Zr, Hf)_5Sb_3(Ni, Cu, Zn)$.

The rare earth antimonides were prepared by melting the components under argon atmosphere and at low excitation voltages in an arc melting furnace. The compositions were controlled by checking the weight losses during the preparation; they were found to be negligible. The ternary Ti, Zr and Hf compounds were prepared by sintering stoichiometric mixtures in evacuated silica tubes for up to eight weeks at 800°C.

Rare earth antimonides

All rare earth compounds could be indexed with hexagonal unit cells and lattice constants as given in Table 1. These compounds crystallize with the normal $D8_8$ structure type: space group $P6_3/mcm$ (D_{3h}^{32}) with four R atoms in $4(d)$, six R atoms in $6(g_I)$ with $x_I \sim 0.25$, and 6Sb in $6(g_{II})$ with $x_{II} \sim 0.61$. The structure of Ho_5Sb_3 has been refined from powder diffractometer tracings, using 46 observed and unobserved reflections and the least-squares program by Gantzel, Sparks & Trueblood (1961). The final parameters are $x_{Ho} = 0.2437 \pm 0.0002$, $x_{Sb} = 0.6085 \pm 0.0003$. The isotropic temperature factor B of Ho was found to be 1.2, and

the isotropic temperature factor B of Sb was calculated to be 0.4.

The R value $\frac{\sum ||F_{obs}| - |F_{calc}||}{\sum |F_{obs}|}$ was found to be 0.097. The intensity calculation for Ho_5Sb_3 using the new parameters is given in Table 2.

Table 2. Intensity calculation for Ho_5Sb_3 with $D8_8$ structure type

| hkl | d_c | $10^3 \sin^2 \theta_c$ | $10^3 \sin^2 \theta_o$ | I_c | I_o |
|----------------|-------|------------------------|------------------------|--------|---------|
| 10 $\bar{1}$ 0 | 7.665 | 10.10 | 10.04 | 45.8 | 45 |
| 11 $\bar{2}$ 0 | 4.425 | 30.29 | 30.15 | 0.5 | < 8 |
| 20 $\bar{2}$ 0 | 3.832 | 40.39 | 40.43 | 64.5 | 22 |
| 11 $\bar{2}$ 1 | 3.608 | 45.56 | 45.74 | 5.8 | < 8 |
| 0002 | 3.116 | 61.07 | 61.01 | 161.3 | 281 |
| 21 $\bar{3}$ 0 | 2.897 | 70.68 | 70.70 | 339.7 | } 471* |
| 10 $\bar{1}$ 2 | 2.887 | 71.17 | 71.42 | 197.0 | |
| 21 $\bar{3}$ 1 | 2.627 | 85.95 | 85.97 | 1000.0 | 1000 |
| 3030 | 2.555 | 90.87 | 90.63 | 451.7 | } 1323* |
| 11 $\bar{2}$ 2 | 2.548 | 91.36 | 91.63 | 782.7 | |
| 20 $\bar{2}$ 2 | 2.418 | 101.46 | 101.42 | 51.0 | 84 |
| 2240 | 2.212 | 121.16 | — | 6.0 | < 8 |
| 3140 | 2.126 | 131.26 | } 131.36 | 31.3 | } 40 |
| 2132 | 2.122 | 131.75 | | 1.4 | |
| 2241 | 2.085 | 136.43 | 136.72 | 61.9 | 71 |
| 3141 | 2.012 | 146.53 | 146.44 | 84.2 | 113 |
| 3032 | 1.976 | 151.94 | 152.05 | 0.8 | < 8 |
| 4040 | 1.916 | 161.55 | 161.86 | 18.2 | 13 |
| 1123 | 1.880 | 167.70 | — | 0.5 | < 8 |
| 2242 | 1.804 | 182.23 | 186.4 | 188.9 | 174 |
| 3240 | 1.758 | 191.84 | } 192.14 | 6.1 | } 29 |
| 3142 | 1.756 | 192.33 | | 26.0 | |
| 3251 | 1.692 | 207.11 | 207.23 | 107.4 | } 340* |
| 2133 | 1.688 | 208.09 | 208.51 | 239.7 | |

Table 1. R_5Sb_3 phases with $D8_8$ structure type

| | a | c | c/a | V | Reference* |
|------------|-------------------|-------------------|--------|-------|------------|
| La_5Sb_3 | 9.42 ± 0.01 | 6.62 ± 0.01 | 0.703 | 508.7 | (1) |
| Ce_5Sb_3 | 9.302 ± 0.009 | 6.514 ± 0.005 | 0.7003 | 488.1 | (1) |
| Pr_5Sb_3 | 9.233 ± 0.004 | 6.510 ± 0.004 | 0.7050 | 480.6 | (1) |
| Nd_5Sb_3 | 9.180 ± 0.005 | 6.463 ± 0.005 | 0.7041 | 471.7 | (1) |
| Gd_5Sb_3 | 8.975 ± 0.004 | 6.343 ± 0.003 | 0.7067 | 442.5 | (1) |
| Tb_5Sb_3 | 8.920 ± 0.003 | 6.304 ± 0.003 | 0.7067 | 434.4 | (1) |
| Dy_5Sb_3 | 8.870 ± 0.004 | 6.266 ± 0.003 | 0.7064 | 427.0 | (1) |
| Ho_5Sb_3 | 8.851 ± 0.002 | 6.234 ± 0.002 | 0.7043 | 423.0 | (1) |
| Yb_5Sb_3 | 8.995 | 6.870 | 0.764 | 481.4 | (2) |

* (1) This work. (2) Bodnar & Steinfink, 1967.

Table 2 (cont.)

| <i>hkl</i> | <i>d_c</i> | 10 ³ sin ² θ _c | 10 ³ sin ² θ ₂ | <i>I_c</i> | <i>oI</i> |
|------------|----------------------|---|---|----------------------|-----------|
| 4150 | 1.672 | 212.03 | 212.49 | 86.8 | 60 |
| 4042 | 1.632 | 222.62 | 222.87 | 123.4 | 83 |
| 4151 | 1.615 | 227.30 | — | 6.1 | < 8 |
| 0004 | 1.558 | 244.29 | 244.43 | 87.3 | 130 |
| 5050 | 1.533 | 252.42 | 252.58 | 33.7 | } 97* |
| 3252 | 1.531 | 252.91 | 253.02 | 34.4 | |
| 1014 | 1.527 | 254.38 | — | 0.9 | < 8 |
| 2243 | 1.514 | 258.57 | 259.07 | 21.0 | 22 |
| 3143 | 1.486 | 268.67 | 268.67 | 30.5 | 45 |
| 3360 | 1.475 | 272.61 | } 273.78 | 4.0 | } 30 |
| 4152 | 1.473 | 273.10 | | 28.7 | |
| 1124 | 1.470 | 274.58 | — | 0.0 | < 8* |
| 4260 | 1.448 | 282.71 | 282.69 | 66.3 | } 65 |
| 2024 | 1.443 | 284.67 | 284.27 | 5.2 | |
| 3361 | 1.435 | 287.88 | 287.90 | 63.1 | 53 |
| 4261 | 1.411 | 297.98 | 298.23 | 28.1 | 32 |
| 5160 | 1.376 | 313.00 | } 313.51 | 25.2 | } 190 |
| 5052 | 1.375 | 313.49 | | 135.4 | |
| 2134 | 1.372 | 314.96 | 315.13 | 55.3 | 72 |
| 5161 | 1.344 | 328.27 | } 328.66 | 48.3 | } 96 |
| 3253 | 1.342 | 329.25 | | 44.8 | |
| 3362 | 1.333 | 333.68 | 334.25 | 45.3 | } 150* |
| 3034 | 1.330 | 335.16 | 335.57 | 97.7 | |
| 4262 | 1.313 | 343.78 | 343.83 | 10.8 | 13 |
| 4153 | 1.303 | 349.44 | — | 2.8 | < 8 |
| 6060 | 1.277 | 363.48 | — | 1.9 | < 8 |
| 2244 | 1.274 | 365.45 | — | 1.5 | < 8 |
| 4370 | 1.260 | 373.58 | — | 2.9 | < 8 |
| 5162 | 1.259 | 374.07 | — | 0.0 | < 8 |
| 3144 | 1.256 | 375.54 | — | 10.4 | 21 |
| 4371 | 1.235 | 388.85 | 389.23 | 15.8 | 17 |
| 5270 | 1.227 | 393.78 | 393.48 | 13.1 | 8 |
| 4044 | 1.209 | 405.83 | — | 5.9 | < 8 |
| 5271 | 1.204 | 409.04 | } 410.09 | 35.2 | } 60 |
| 3363 | 1.202 | 410.02 | | 32.4 | |

* For these line pairs it was possible to measure the individual θ values but only a common intensity value.

In Fig. 1 are plotted the cube roots of the volume of one formula unit R_5Sb_3 versus the 3-valent ionic radii of the rare earth elements in a way which we have discussed before (Hohnke & Parthé, 1966; Parthé, 1967). The slope of the data points indicates that the electronic state of the rare earth metals in these compounds is the same except for Yb which shows an extraordinarily big deviation. The deviation is also

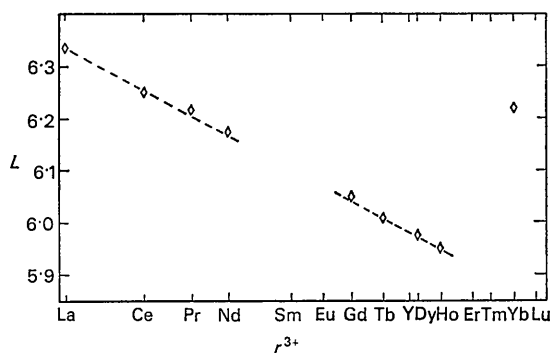


Fig. 1. Cube roots of the formula units of R_5Sb_3 compounds with $D8_8$ structure type versus the 3-valent ionic radii of the rare earth metals.

found with the axial ratio as can be seen in Table 1, where the c/a index for Yb_5Sb_3 is 0.764, whereas the c/a ratio for the other compounds is ~ 0.707 .

Ni, Cu- or Zn-stabilized antimonides of Ti, Zr and Hf

All these compounds could again be indexed with hexagonal unit cells, and lattice constants as given in Table 3. The extinctions are the same as for the $D8_8$ structure type ($h0\bar{h}l$ with $l=2n+1$); however, the structure is now a filled up version of the $D8_8$ structure type as described for Hf_5Sn_3Cu (Rieger, Nowotny & Benesovsky, 1965): space group $P6_3/mcm$ (D_{6h}^{3h}) with 4 Hf in $4(d)$, 6 Hf in $6(g_{II})$ with $x_{II} \sim 0.27$, 6 Sb in $6(g_{II})$ with $x_{II} \sim 0.61$, 2 Cu in $2(b)$. An intensity calculation using these parameters was made for Hf_5Sb_3Zn as shown in Table 4. The agreement is excellent; it was not felt necessary to perform a refinement on this structure.

Table 3. Ternary antimonides with Hf_5Sn_3Cu type, a filled-up version of the $D8_8$ structure type

| | <i>a</i> | <i>c</i> | <i>c/a</i> | <i>V</i> |
|--------------|-------------------|-------------------|------------|----------|
| Ti_5Sb_3Cu | 8.020 ± 0.010 | 5.515 ± 0.010 | 0.687 | 307.2 |
| Zr_5Sb_3Ni | 8.530 ± 0.005 | 5.773 ± 0.005 | 0.677 | 363.8 |
| Zr_5Sb_3Cu | 8.526 ± 0.007 | 5.786 ± 0.005 | 0.679 | 364.3 |
| Zr_5Sb_3Zn | 8.555 ± 0.007 | 5.820 ± 0.005 | 0.680 | 368.9 |
| Hf_5Sb_3Ni | 8.469 ± 0.004 | 5.715 ± 0.004 | 0.675 | 355.0 |
| Hf_5Sb_3Cu | 8.476 ± 0.004 | 5.724 ± 0.004 | 0.675 | 356.1 |
| Hf_5Sb_3Zn | 8.514 ± 0.003 | 5.747 ± 0.003 | 0.675 | 360.8 |

Table 4. Intensity calculation for Hf_5Sb_3Zn with Hf_5Sn_3Cu structure type

Cr $K\alpha$ radiation. $x_{Hf} = 0.270$, $x_{Sb} = 0.610$.

| <i>hkl</i> | <i>d_c</i> | 10 ³ sin ² θ _c | 10 ³ sin ² θ _o | <i>I_c</i> | <i>I_o</i> |
|------------|----------------------|---|---|----------------------|----------------------|
| 1010 | 7.3733 | 24.13 | — | 21 | — |
| 1110 | 4.2570 | 72.40 | — | 6 | — |
| 2020 | 3.6867 | 96.54 | — | 28 | — |
| 1121 | 3.4208 | 112.13 | 113.6 | 60 | <i>vw</i> |
| 0002 | 2.8735 | 158.90 | 160.3 | 102 | <i>vw</i> |
| 2130 | 2.7869 | 168.94 | 170.7 | 153 | <i>w</i> |
| 1012 | 2.6774 | 183.04 | 184.0 | 67 | <i>vw</i> |
| 2131 | 2.5076 | 208.67 | 210.4 | 719 | <i>s</i> |
| 3030 | 2.4578 | 217.21 | 219.0 | 801 | <i>s+</i> |
| 1122 | 2.3817 | 231.31 | 232.8 | 1000 | <i>vs</i> |
| 2022 | 2.2664 | 255.44 | — | 16 | — |
| 2240 | 2.1285 | 289.61 | — | 15 | — |
| 3140 | 2.0450 | 313.74 | — | 3 | — |
| 2132 | 2.0005 | 327.84 | — | 1 | — |
| 2241 | 1.9960 | 329.34 | — | < 1 | — |
| 3141 | 1.9267 | 353.47 | 355.5 | 149 | <i>m</i> |
| 3032 | 1.8678 | 376.11 | — | 11 | — |
| 4040 | 1.8433 | 386.15 | — | 26 | — |
| 1123 | 1.7469 | 429.94 | — | 9 | — |
| 2242 | 1.7104 | 448.51 | 449.5 | 272 | <i>s-</i> |
| 3250 | 1.6916 | 458.55 | — | 9 | — |
| 3142 | 1.6661 | 472.65 | 473.0 | 30 | <i>vw</i> |
| 3251 | 1.6227 | 498.27 | 499.1 | 152 | <i>m+</i> |
| 4150 | 1.6090 | 506.82 | 507.7 | 68 | <i>wm</i> |
| 2133 | 1.5787 | 526.47 | 527.5 | 201 | <i>ms</i> |
| 4042 | 1.5515 | 545.05 | 546.2 | 123 | <i>m</i> |
| 4151 | 1.5494 | 546.54 | — | 2 | — |
| 5050 | 1.4747 | 603.35 | — | 10 | — |
| 3252 | 1.4577 | 617.45 | — | 8 | — |
| 0004 | 1.4367 | 635.62 | 636.1 | 137 | <i>m+</i> |
| 2243 | 1.4239 | 647.15 | — | < 1 | — |
| 3360 | 1.4190 | 651.62 | — | 53 | <i>w</i> |

Table 4 (cont.)

| <i>hkl</i> | e_e | $10^3 \sin^2 \theta_e$ | $10^3 \sin^2 \theta_o$ | I_c | I_o |
|------------|--------|------------------------|------------------------|-------|------------|
| 1014 | 1.4102 | 659.75 | — | 1 | — |
| 4152 | 1.4039 | 665.72 | 666.1 | 223 | <i>s</i> — |
| 3143 | 1.3981 | 671.28 | 671.0 | 87 | <i>m</i> |
| 4260 | 1.3934 | 675.76 | 675.9 | 82 | <i>m</i> |
| 3361 | 1.3776 | 691.35 | 691.4 | 49 | <i>w</i> |
| 1124 | 1.3613 | 708.02 | — | <1 | — |
| 4261 | 1.3542 | 715.48 | 716.0 | 183 | <i>ms</i> |
| 2024 | 1.3387 | 732.16 | 732.2 | 6 | <i>vw</i> |
| 5160 | 1.3243 | 748.16 | 748.5 | 89 | <i>m</i> |
| 5052 | 1.3120 | 762.26 | 762.7 | 99 | <i>m</i> |
| 5161 | 1.2905 | 787.88 | 788.0 | 289 | <i>s</i> |
| 2134 | 1.2770 | 804.56 | 805.0 | 77 | <i>wm</i> |
| 3362 | 1.2723 | 810.53 | 811.0 | 81 | <i>wm</i> |
| 3253 | 1.2680 | 816.08 | 816.0 | 174 | <i>ms</i> |
| 4262 | 1.2538 | 834.66 | 834.6 | 44 | <i>vw</i> |
| 3034 | 1.2404 | 852.83 | 852.3 | 633 | <i>vs</i> |
| 4153 | 1.2321 | 864.35 | — | 3 | — |
| 6060 | 1.2289 | 868.83 | 868.6 | 39 | <i>vw</i> |
| 4370 | 1.2122 | 892.96 | — | 25 | — |
| 5162 | 1.2027 | 907.06 | 906.5 | 59 | <i>vw</i> |
| 2244 | 1.1908 | 925.23 | 925.5 | 26 | <i>vw</i> |
| 4371 | 1.1861 | 932.69 | 932.9 | 43 | <i>vw</i> |
| 5270 | 1.1807 | 941.23 | 940.8 | 177 | <i>m</i> |

The similarity of the X-ray patterns indicates that all the other compounds listed in Table 3 were of the same structure type.

Discussion

Stable binary antimonides with $D8_8$ structure seem to form only with the rare earth elements. The earlier reported Zr_5Sb_3 (Boller & Parthé, 1963) is extremely unstable; it decomposes in air within half an hour. It seems that Ti_5Sb_3 and Hf_5Sb_3 do not form at all. However, all these fourth-group antimonides can be stabilized by addition of Ni, Cu or even Zn atoms which then occupy octahedral voids in the structure. Our experiments indicate that these three elements are the

only ones to stabilize the $D8_8$ structure with the fourth-group antimonides. In particular, it was found impossible to use C, B, Fe, Co, Mn, Ru, Pt or Ir as stabilizers. In this connection, it should be mentioned that V_5As_3 does not exist as binary $D8_8$ phase but must be stabilized by carbon (Boller & Nowotny, 1966). Whether a particular stabilizing atom can be used or not depends, it seems, upon the size of the octahedral void.

A detailed discussion about the occurrence of all representatives of the $D8_8$ and related structure types will be given at a later time.

This study is a contribution from the Laboratory for Research on the Structure of Matter, University of Pennsylvania, supported by the Advanced Research Projects Agency, Office of the Secretary of Defense. It was supported by the U.S. Public Health Service under USPH-grant DE-02135. The financial assistance of the sponsoring agency is gratefully acknowledged.

References

- BODNAR, R. E. & STEINFINK, H. (1967). *Inorg. Chem.* **6**, 327.
 BOLLER, H. & NOWOTNY, H. (1966). *Mh. Chem.* **97**, 1053.
 BOLLER, H. & PARTHÉ, E. (1963). *Mh. Chem.* **94**, 225.
 GANTZEL, P. K., SPARKS, R. A. & TRUEBLOOD, K. N. (1961). Program UCLA-LS 1.
 HOHNKE, D. & PARTHÉ, E. (1966). *Acta Cryst.* **21**, 435.
 JEITSCHKO, W. & PARTHÉ, E. (1965). *Acta Cryst.* **19**, 275.
 JEITSCHKO, W. & PARTHÉ, E. (1967). *Acta Cryst.* **22**, 551.
 PARTHÉ, E. (1967). In *Propriétés Thermodynamiques, Physiques et Structurales des Dérivés Semi-Metalliques*. Colloques Internationaux du Centre National de la Recherche Scientifique, No. 157, pages 195–205. Paris: Editions du CNRS.
 RIEGER, W., NOWOTNY, H. & BENESOVSKY, F. (1965). *Mh. Chem.* **96**, 98, 232.
 SCHOB, O. & PARTHÉ, E. (1964). *Acta Cryst.* **17**, 1335.